Controlled antiseptic release by alginate polymer films and beads

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ABSTRACT

Biodegradable polymeric materials based on blending aqueous dispersions of natural polymer sodium alginate (NaAlg) and povidone iodine (PVPI) complex, which allow controlled antiseptic release, are presented. The developed materials are either free standing NaAlg films or Ca²⁺-cross-linked alginate beads, which properly combined with PVPI demonstrate antibacterial and antifungal activity, suitable for therapeutic applications, such as wound dressing. Glycerol was used as the plasticizing agent. Film morphology was studied by optical and atomic force microscopy. It was found that PVPI complex forms well dispersed circular micro-domains within the NaAlg matrix. The beads were fabricated by drop-wise immersion of NaAlg/PVPI/glycerol solutions into aqueous calcium chloride solutions to form calcium alginate beads encapsulating PVPI solution (CaAlg/PVPI). Controlled release of PVPI was possible when the composite films and beads were brought into direct contact with water or with moist media. Bactericidal and fungicidal properties of the materials were tested against *Escherichia coli* bacteria and *Candida albicans* fungi. The results indicated very efficient antibacterial and antifungal activity within 48 h. Controlled release of PVPI into open wounds is highly desired in clinical applications to avoid toxic doses of iodine absorption by the wound. A wide variety of applications are envisioned such as external and internal wound dressings with controlled antiseptic release, hygienic and protective packaging films for medical devices, and polymer beads as water disinfectants.

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1. Introduction

Naturally occurring alginate polymers have long been used in the food and beverage industries as thickening, gel-forming and colloidal stabilizing agents (Donati & Paoletti, 2009; Moe, Dragnet, Skjak-Bræk, & Smidsrod, 1995). They are also used in the pharmaceutical industry as matrices for drug encapsulation, as substrates for cell culture, as binders for medical tablets (Goh, Heng, & Chan, 2012) and for many applications of controlled transdermal or trans-mucosal drug delivery of active substances (Sachan, Pushkar, Jha, & Bhattacharya, 2009). Commercial alginates are extracted primarily from marine algae such as *Laminaria hyperborea*, *Ascophyllum nodosum* and *Macrocystis pyrifera* (Wee & Gombotz, 1998). Alginates are linear unbranched polysaccharides containing varying proportions of beta-D-mannuronate (M) and alpha-L-guluronate

(G) residues. Molecular variability in this polymer depends on the source of marine algae, tissue from which alginates are extracted, and also the season of crop harvesting. The composition, sequence of polymer blocks and molecular weight of alginates are important as these factors determine the physical properties of the gel formed. In solution, alginates behave like flexible coils, but when they interact with calcium ions (such as in a calcium chloride solution) they form ordered structures. In fact, cross linking of water soluble sodium alginate can be initiated by polyvalent cations such as calcium, forming water insoluble calcium alginate. The mechanism of CaAlg bead formation involves cooperative binding of calcium ions between two aligned GG blocks of two alginate chains. The formation of CaAlg beads is an instantaneous and irreversible process, which is determined by the rate of diffusion of calcium ions in the sodium alginate polymer matrix (Blandino, Macias, & Cantero, 1999).

Recent trends in biomedical and pharmaceutical industry indicate that alginates (Goh et al., 2012; Shilpa, Agrawal, & Ray, 2003), are increasingly explored for the development of new drug delivery materials. Among alginates, sodium alginate (NaAlg) is the most widely explored. Being non-toxic hydrophilic and biodegradable, it can be absorbed from skin into the body fluids without any toxic effects (Sachan et al., 2009), ideal characteristics for the

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wound healing process (Chiu, Lee, Chu, Chang, & Wang, 2008). It has been reported that the potential combination with antimicrobial and enzymatic components can promote elimination of necrotic tissues and microbial bodies, while the polysaccharide base can stimulate reparative wound process (Patel et al., 2007). Chitosan is another natural marine polymer extracted from sea crustaceans with exceptional properties (Paños, Acosta, & Heras, 2008; Spano, Massaro, Blasi, Malerba, & Cingolani, 2012). The combination of NaAlg with chitosan has become quite commonplace for the development of potential wound healing materials (films, and membranes) as they showed no toxic effects to mammalian cells (Wang, Khor, Wee, & Yong Lim, 2002). A very recent study has shown that NaAlg/chitosan-based wound dressing films in combination with a special laser treatment therapy significantly accelerated burn healing by modulating the epithelization, blood vessels formation and collagenization process (Dantas et al., 2011). Extract of plants that have wound healing properties have also been incorporated into alginate films. Asiaticoside, a component derived from *Centella asiatica* Linn., has been used with alginate to produce films to heal wounds, burns and ulcer abnormalities on the skin (Sikareepaisana, Ruktanonchai, & Supaphol, 2011). Alginate and alginate/chitosan microcapsules with controlled drying properties have shown to increase the viability of the encapsulated probiotic *Bifidobacterium breve* during exposure at stimulated gastric fluid (Cook, Tzortzis, Charalampopoulos, & Khutoryanskiy, 2011).

In this work, we present fabrication and characterization of alternative and highly effective antibacterial and antifungal biomaterials combining sodium and calcium alginates with a traditional antiseptic, the povidone iodine complex (PVPI, or betadine). PVPI is a water soluble complex of iodine and polyvinylpyrrolidone, widely used for its bactericidal and fungicidal properties. PVPI is in fact an iodophor. Iodophors are compounds of iodine linked to surfactants that act as carriers or solubilizing agents. As such, a small amount of free iodine is released in solution, thereby minimizing toxicity while preserving moderate germicidal activity of the element (Heiner, Hile, Demons, & Wedmore, 2010). The most common form of PVPI complex is its aqueous solution available in squeeze bottles containing from 9 to 12 wt% available iodine, which can be purchased off the shelf without prescription. Aqueous PVPI solutions are directly applied on open wounds or infected regions, a process commonly known as wound irrigation. Although slow release of iodine from the PVPI complex in solution minimizes iodine toxicity toward mammalian cells, its sudden absorption by some open wounds during irrigation can cause severe toxicity problems and severe skin burn limiting its surgical applications, particularly under repeated applications (Nahliel, Baruchinb, Levai, Shapiraa, & Yoffea, 2001; Wong et al., 2011). In addition to its wide use for skin antiseptics, PVPI is also employed internally. For instance, it is often applied on large internal wounds, deep tissues or mucosa (Lakhal, Faidherbe, Choukhi, Boissier, & Capdevila, 2011). Due to the ability of polyvinylpyrrolidone polymer to absorb water up to 40% of its weight, PVPI complex films easily lose their consistency or wash away quickly in highly humid or water containing media. In such cases their antibacterial effect is drastically reduced. To avoid this effect, recent studies propose, for instance, immobilization of PVPI complex on polymer surfaces on single-walled carbon nanotube surfaces (Simmons et al., 2009) or incorporation into hydrogels (Mishra & Chaudhary, 2010). We demonstrate a straightforward way to prepare dry NaAlg/PVPI composite films as potential wound dressing materials by simple solution casting from water suspensions, with the ability of controlled release of PVPI into liquid media. Alternatively, by dripping small successive quantities of aqueous solutions of NaAlg and PVPI into a CaCl_2 water solution through a syringe, PVPI solution can be encapsulated in calcium cross linked alginate beads, to enable longer lasting controlled release of PVPI

Fig. 1. Photographs of (a) NaAlg/PVPI complex composite film plasticized with glycerol and (b) CaAlg beads encapsulating PVPI complex liquid.

into liquid media. Both NaAlg/PVPI films and CaAlg/PVPI beads showed highly efficient antimicrobial and antifungal properties against *Escherichia coli* bacteria and *Candida albicans* fungi. These films and beads can be easily adapted to large area dry wounds as dressing, sterile biodegradable surface coatings and to internal wound treatment for slow drug release (Athanasios et al., 2012).

2. Material and method

2.1. Materials

Alginic acid sodium salt with viscosity 15,000–20,000 centipoises (cps), glycerol ≥ 99.5 (GC) and calcium chloride anhydrous were purchased from Sigma–Aldrich. PVPI, 100 ml solution in water contained 10 wt% PVPI of which 1 wt% was active iodine, named as LH IODO 10, was purchased from Pharmatek PMC, Italy.

2.2. Films preparation

Three different NaAlg–PVPI blend films were studied herein. Films were cast from three identical NaAlg in water stock solutions containing different amounts of PVPI solution with a total volume of 10 ml. More specifically, distilled water and 0.3 g of NaAlg were mixed together and warmed-up using a hot plate for 1 h near 100°C to ensure complete dissolution of the alginate in water. Afterwards, the viscous solutions were left to cool to room temperature and to these 0.1 ml of glycerol was added as a plasticizer. The amount of added glycerol was kept constant in all the subsequent solution preparations. Next, 1.0, 1.5 and 2.0 ml of aqueous PVPI solution were slowly blended into the NaAlg/glycerol solutions. The final solutions were mixed using a vortex mixer and left for a while to ensure that no bubbles were present in the vials. For the films' formation, 200 μl of the prepared solutions were drop cast on glass slides of $1\text{ cm} \times 1\text{ cm}$ area and left under ambient laboratory conditions for drying. The resultant dry films obtained from these solutions have the following composition: *Film 1*: 57% NaAlg, 24% glycerol and 19% PVPI; *Film 2*: 52% NaAlg, 22% glycerol and 26% PVPI; *Film 3*: 48% NaAlg, 20% glycerol and 32% PVPI. According to the manufacturer, the specific weight of PVPI polymer is approximately 1.03. Therefore, for all practical purposes, the density of PVPI was taken as 1.0 g/ml for dry film composition calculations. The ratio of NaAlg to glycerol in all the films was between 2.3 and 2.4. Fig. 1a shows a photograph of a representative flexible biodegradable NaAlg/PVPI film that was previously cast on a plastic Petri dish. Samples for Fourier transform infrared (FTIR) spectroscopy analysis were instead prepared by spin coating of 1 ml of the above described solutions (in 2 steps: 500 rpm for 10 s and 750 rpm for 30 s) on silicon wafers and then slightly baked on a hot plate (60°C for 1–2 min) to obtain thin films and avoid saturation effects during the infrared spectroscopy analysis.

2.3. Beads preparation

The above mentioned solution used in the fabrication of *Film 3* (48% NaAlg, 20% glycerol and 32% PVPI), was chosen to prepare CaAlg/PVPI beads. The NaAlg/PVPI/glycerol solution was drop-wise immersed into a 10% CaCl₂ solution in distilled water. The formation of the beads was immediate. Subsequently, the beads were collected and rinsed with distilled water to remove excess CaCl₂ traces. Fig. 1b shows a photograph of CaAlg/PVPI beads collected from the CaCl₂ solution.

3. Experimental

3.1. Optical and atomic force microscopy

Optical microscopy images were acquired with a Zeiss AXIO instrument. For atomic force microscopy (AFM) images, a Park System AFM instrument (XE-100) was used in true non-contact mode, on an anti-vibration table (Table Stable TS-150) and within an acoustic enclosure. Single-beam silicon cantilevers tips (PPP-NCHR-10) were used for the data acquisition with about less than 10 nm nominal radius and 42 N/m elastic force constant for high sensitivity. The resonance frequency was defined around 280 kHz. The scan rate was between 0.2 and 1.0 Hz.

3.2. Fourier transform infrared (FTIR) spectroscopy

FTIR analyses were performed with a VERTEX 70 FTIR instrument in the spectral range from 4000 to 400 cm⁻¹ in absorbance mode.

3.3. Measurement of PVPI release from NaAlg films and CaAlg beads

Film 1 and *Film 3* were chosen for the measurement of PVPI release into distilled water. Small samples cut from the films were placed at the bottom of UV cuvettes, which were then positioned in a CARY 300 Scan UV–visible spectrophotometer from Varian. The cuvettes were filled with 2.5 ml distilled water and the release of PVPI was monitored in time by following the increase of the PVPI characteristic UV–vis absorption peaks. Gentle agitation of the cuvette before each UV–vis absorption measurement step was necessary in order to disperse the released PVPI evenly in the cuvette.

In order to measure the release of PVPI solution from freshly prepared CaAlg beads, 0.5 ml stock solution used to make *Film 3* was converted into beads by drop-wise immersion into CaCl₂ solution. After collection and rinsing, the beads were added into the cuvette containing 2.5 ml distilled water and the characteristic UV–vis absorption peak of PVPI was monitored in time as described above.

3.4. Antibacterial/antifungal activity of NaAlg/PVPI films and CaAlg/PVPI beads

In these experiments, the activity of NaAlg/PVPI films coated on glass slides and CaAlg/PVPI beads was tested against two different organisms, namely *E. coli*, a model Gram-negative bacterium, and *C. albicans*, a fungus which is the causal agent of many opportunistic oral and genital infections in humans.

A loop of glycerol stock of *E. coli* strain TG1 was streaked onto a Luria–Bertani (LB) medium agar plate and incubated overnight at 37 °C. Then a single colony was picked up and grown in LB liquid medium overnight at 37 °C to reach a bacterial concentration of $\sim 8 \times 10^8$ cells/ml, in a shaking incubator. To prepare the fungal cell suspension, two to three *C. albicans* colonies were picked from the plate, washed in isotonic glucose phosphate buffer (IGP; 1 mM

Fig. 2. (a) Representative optical microscopy image showing the morphology of 70 wt% NaAlg and 30 wt% glycerol dry films, drop cast from a water solution. (b) Representative optical microscopy image showing the morphology of *Film 3* (48% NaAlg, 20% glycerol and 32% PVPI). Micro-phase separation is evident from the appearance of micro-circular features throughout the sample.

potassium phosphate buffer, pH 7.0, supplemented with 287 mM glucose as an osmoprotectant), and a fungal cell suspension was prepared by adjusting the inoculum to 10⁸ cells/ml.

To assess the antibacterial behavior of NaAlg/PVPI films, the overnight *E. coli* culture was diluted to a final concentration of 10⁶ cells/ml, and 500 μl of this solution was spread onto new LB medium agar plates. The plates were then placed in the incubator at 37 °C for 2 h (to allow a proper evaporation of residual liquid medium), followed by the positioning of the different substrates (2 for each LB plates). To investigate the antifungal activity of NaAlg/PVPI films, the initial 10⁸ cells/ml inoculum was diluted to 10⁵ cells/ml and 100 μl of this solution was spread onto freshly prepared Sabouraud dextrose agar (SDA). The plates were then incubated at 37 °C for 2 h, followed by the positioning of the different substrates (2 for each plate). The substrates were overnight sterilized, by UV irradiation, inside a biohazard hood before their use.

The antibacterial/antifungal behavior of CaAlg/PVPI beads was tested by turbidity assay. In particular, a confluent overnight *C. albicans* culture was diluted to an optical density (OD₆₀₀) of 0.1, then 1 g of CaAlg/PVPI beads was added to the culture and the optical density, at 600 nm the commonly used wavelength for cell growth evaluation, was measured every 30 min with a spectrophotometer (Thermo Scientific). The same turbidity assays were performed with *E. coli*, but in this case 12 g of CaAlg/PVPI beads were supplemented to the bacteria culture. All the beads used for the experiments were previously sterilized by an overnight UV irradiation inside a biohazard hood.

4. Results and discussion

4.1. Investigation of the NaAlg/PVPI films morphology with optical and atomic force microscopy

Fig. 2(a) illustrates a representative optical microscope image showing the morphology of a glycerol plasticized NaAlg dry film made up of 70 wt% NaAlg and 30 wt% glycerol. Fig. 2(b), on the other hand, shows the optical microscope image of a dry *Film 3* (48% NaAlg, 20% glycerol and 32% PVPI). Careful inspection of Fig. 2(b) reveals the existence of circular domains with 10–20 μm diameter throughout the composite film. This is a first indication that PVPI and NaAlg are not fully miscible, and micro-phase separation occurs during drying of the composite films.

Fig. 3 shows images of AFM surface topography and phase contrast of different drop cast films. The films were drop cast on thin glass slides from their respective stock solutions. The dry film shown in Fig. 3(a) contains 100 wt% NaAlg, whereas the dry film in Fig. 3(b) contains 70 wt% NaAlg and 30 wt% glycerol. Glycerol is

Fig. 3. AFM topographical (denoted by 1) and phase (denoted by 2) images ($40\ \mu\text{m} \times 40\ \mu\text{m}$) of (a) a dry NaAlg film; (b) 70 wt% NaAlg and 30 wt% glycerol; (c) Film 1: 57% NaAlg, 24% glycerol and 19% PVPI; (d) Film 2: 52% NaAlg, 22% glycerol and 26% PVPI and (e) Film 3: 48% NaAlg, 20% glycerol and 32% PVPI.

a low-molecular weight nonvolatile substance and is a common plasticizer for natural polymers and pharmaceutical materials. It was shown to reduce internal hydrogen bonding between polymer chains while increasing molecular volume, which results in film flexibility (Maizura, Fazilah, Norziah, & Karim, 2007). Maizura et al. also demonstrated that NaAlg films plasticized by glycerol show highly smooth and defect (crack) free surface morphology, whereas pure solution cast NaAlg films can form cracks at times reducing their mechanical properties as well as increasing their water vapor permeability. Moreover, the Young's modulus of NaAlg films is reduced and their percent elongation at break is increased with the incorporation of glycerol (Russo, Malinconico, Petti, & Romano, 2005), which is in accordance with the Dynamic Mechanical Analysis measurements conducted on our films (not shown here). Morphological and phase AFM analysis results from the PVPI-incorporating Films 1, 2 and 3 are shown in Fig. 3(c, d and e), respectively. Various sizes of micro-circular domains attributed to PVPI are clearly seen in the composite structures. It is also observed that for increasing PVPI concentration the circular domains inside the NaAlg–glycerol matrix become larger. From the phase images, it is clear that the PVPI circular domains are well dispersed and separated from the matrix.

The results are in accordance with a recent work (Caykara, Demirci, Eroglu, & Guven, 2006) on NaAlg/polyvinylpyrrolidone (PVP) (without iodine) blend films. Various AFM results from this study also indicated phase separation in the form of circular PVP micro domains in NaAlg films. Their analysis based on acid–base adhesion theory indicated that the surface free energy of PVP has a basic character whereas the surface free energy of NaAlg is somewhat acidic. This led to the conclusion that when the surface acidity of one component is high while the surface basicity of the other component is also high, the two components show good compatibility, though not miscible, caused by strong adhesion between them, even if micro-phase separation takes place.

4.2. Fourier transform infrared spectroscopy

In Fig. 4, the FTIR spectra of all the fabricated films obtained in the range $400\text{--}4000\text{ cm}^{-1}$ wave numbers are shown. The main peak, common in all the films, at 3380 cm^{-1} is attributed to the OH stretching, which is readily present in the alginic acid backbone of NaAlg, and to the presence of $(\text{O} \cdots \text{H}^+ \cdots \text{O})$ bridges within the PVPI complex film. In all the samples, there is also a peak

at 2943 cm^{-1} , which is attributed to CH_2 stretching. The peak at 2890 cm^{-1} appears in samples plasticized with Glycerol, corresponding to spectra denoted by (1) to (5), and is attributed to CH band stretching. The same peak is also present in the PVPI spectrum, denoted by spectra (6), and originates from the PVP backbone. Detailed inspection of spectra also indicates existence of a shoulder at 1463 cm^{-1} , which is present in samples plasticized with glycerol and can be attributed to the C–O band stretching of the primary alcohol. The other clearly distinguished peaks appearing in all the six samples are the anti-symmetric carboxylate band stretching centered around 1596 cm^{-1} and the symmetric carboxylate band stretching around 1420 cm^{-1} , and were also observed in previous studies on NaAlg polymers (Lawrie et al., 2007; Sartori, Finch, Ralph, & Gilding, 1997). In addition, the peak at 1665 cm^{-1} was only present in the composites with PVPI, namely the spectra denoted

Fig. 4. FTIR spectra of all the films studied: (1) 100 wt% NaAlg film; (2) 70 wt% NaAlg, 30 wt% glycerol (plasticizer) film; (3) Film 1: 57% NaAlg, 24% glycerol and 19% PVPI; (4) Film 2: 52% NaAlg, 22% glycerol and 26% PVPI; (5) Film 3: 48% NaAlg, 20% glycerol and 32% PVPI and (6) pure PVPI complex film.

Fig. 5. UV absorption curves as a result of PVPI release from (a) 9.0 mg of *Film 1*, (b) 5.0 mg of *Film 3*, (c) CaAlg beads formed from 0.5 ml of solution with the highest PVPI amount and (d) PVPI release kinetics in 2.5 ml water.

by (3), (4), (5) and (6) and is attributed to the C=O band stretching in PVPI complex. The shoulder at 1323 cm^{-1} was also only present in samples with PVPI and is due to aromatic amines present in the PVP molecule. Furthermore, the peak at 1296 cm^{-1} is due to the C–O stretch of the samples containing PVPI molecules. The peaks at 1100 and 1040 cm^{-1} are due to C–O–C anti-symmetric stretch and C–O skeletal vibration of C–O stretching (Kongjao, Damronglerd, & Hunsom, 2010). We also observed that the higher the amount of the PVPI in NaAlg films, the higher the ratio of the intensity of PVPI associated carboxylate peak (1665 cm^{-1}) to the NaAlg associated 1596 cm^{-1} carboxylate peak. In summary, the FTIR analysis indicates that there is no formation of new species when NaAlg and PVPI are blended in solution. This is an important conclusion since it demonstrates that upon physical mixing of the two polymers, NaAlg does not deteriorate or degrade the chemical constitution of the PVPI complex, so that PVPI can serve its purpose as a drug when embedded inside the NaAlg polymer matrix.

4.3. Measurement of PVPI release from NaAlg films and CaAlg beads

The dynamic release of the PVPI from the NaAlg films in water was investigated using UV absorption spectroscopy. Pieces of 9.0 mg of *Film 1* (18 wt% PVPI) and 5.0 mg of *Film 3* (32 wt% PVPI), containing overall the same amount of PVPI, were cut and placed at the bottom of different UV cuvettes containing 2.5 ml water each. Their UV absorption was recorded from the moment of their water immersion until the termination of release of the PVPI. According to literature, PVPI has two distinct absorption

peaks due to the complexed I_3^- which are around 288 and 351 nm (Mazumdar, Chikindas, & Ulrich, 2010). PVPI containing NaAlg films are expected to start swelling and then dissolve in water, releasing PVPI until the dissolution is completed. The measured absorbance intensity increases with time as more and more PVPI is released into water due to the swelling and subsequent dissolution of the NaAlg matrix (Fig. 5a and b). A small red shift of the two absorption peaks, and especially of the one at longer wavelengths, is noted, possibly due to the electron density changes caused by the complexation changes of the iodine ions and/or to conformational changes in the PVP polymer's backbone after its dissolution in water. The intensity of the second absorption peak was used for the evaluation of the PVPI release in water. The release dynamics are similar for the two films studied (first graph Fig. 5(d)) indicating that the PVPI concentration in the alginate films is not affecting its release. The absorption increase rate starts reducing after 7 min, indicating approach of PVPI release termination. Within 11–13 min, the absorption intensity no longer increases indicating maximum PVPI release. Therefore, about 12 min are enough for the complete release in 2.5 ml of water of a total amount of 1.60 mg PVPI incorporated in NaAlg films, independently of the percentage concentration of the complex in the polymeric films.

Additionally, the release rate of encapsulated PVPI solution into freshly prepared CaAlg beads was also investigated in water. Beads obtained as described in Section 2.3, from approximately 0.5 g of the solution used for the preparation of *Film 3* were immersed in the same cuvette used for the films, containing 2.5 ml distilled water for UV absorption measurements. Fig. 5(c) shows the characteristic absorption spectra of the PVPI complex. It should be noted that

Fig. 6. (a) Beads, obtained from 0.5 ml of *Film 3* solution, placed inside a vial containing 2.5 ml of water after 4 h, (b) beads, obtained from 1 ml of *Film 3* solution, placed inside a vial containing 2.5 ml of water after 4 h and (c) 0.5 ml of *Film 3* solution directly dispersed in 2.5 ml water.

the red shift of the absorption peaks is quite reduced compared to the one monitored during the films' release. This indicates that the PVPI that is encapsulated in the CaAlg beads is already dissolved in water, and therefore the complexation of the iodine and/or the conformation of the PVP polymer's backbone should remain similar after its release. The intensity of the second absorption peak was used also in this case for the evaluation of the PVPI release in water. The release dynamics are represented in the second graph of Fig. 5(d), which shows that the release of PVPI in 2.5 ml of water reaches a full plateau within 4 h. The total amount of PVPI in the beads was calculated to be approximately 10 mg (density of PVPI was taken as 1.0 g/ml). Note that the maximum absorption intensity of the PVPI released from the dissolving films is approximately twice than that released from the beads, even though the beads contain much more PVPI (10 mg) than the NaAlg films (1.60 mg). This observation indicates that at the end of the release procedure some PVPI should remain entrapped inside the CaAlg beads. To verify this assumption we performed the experiment presented in Fig. 6. In particular, Fig. 6(a) shows a photograph of CaAlg beads prepared

from 0.5 ml *Film 3* solution, 4 h after they were placed inside a vial containing 2.5 ml water. Fig. 6(b), on the other hand, shows the state of the beads, which were obtained from 1 ml of *Film 3* solution, after 4 h in 2.5 ml water. Finally, Fig. 6(c) shows a photograph of the 0.5 ml *Film 3* solution directly dispersed in 2.5 ml of distilled water. Qualitative comparison of Fig. 6(a) and (c) indicates that the PVPI released from CaAlg beads in water after 4 h is clearly less than the one directly diluted in water using the same solution in amount and composition. This is due to some PVPI, which remains encapsulated inside the beads as clearly seen in Fig. 6(a). Even when the quantity of the beads, and thus the quantity of the entrapped PVPI, is doubled (Fig. 6b), there still appears to be less dissolved PVPI in the water compared to the one of the *Film 3* solution directly dispersed in 2.5 ml water. We propose that although the CaAlg bead formation is instant upon immersion of the NaAlg/PVPI droplets into the CaCl_2 solution, it is possible that some of the PVPI may directly leach into the CaCl_2 solution before or during the formation (cross linking) of the beads reducing the amount of encapsulated PVPI in the beads. Moreover, since iodine is complexed into PVP as I_3^- , calcium ions can easily form embedded CaI_2 structures during alginate cross linking. In fact, this can be easily observed if some PVPI solution is added drop wise directly into excess CaCl_2 solution, where the characteristic orange color of PVPI slowly turns colorless. CaI_2 is a standard medicinal compound which is especially effective for bronchopulmonary diseases having also a partial anti-septic effect (Lopez-Belio, Henderson, Mansueto, & Holinger, 1961). Therefore, formation of CaI_2 is not considered as a drawback but as an added functionality opening up new application possibilities for these beads such as orally administered antiseptic and expectorant (cough, asthma, bronchitis and cystic fibrosis relieving agent). CaI_2 is also an excellent dietary supplement (Berasategi et al., 2010).

4.4. Antibacterial/antifungal activity of NaAlg/PVPI films and CaAlg/PVPI beads

The antibacterial and antifungal activity of the composite films against *E. coli* and *C. albicans* was tested on films obtained by drop casting on two identical small glass slides which were placed

Fig. 7. Photographs of Petri dishes containing heavily populated *E. coli* bacteria after 48 h incubation in the presence of (a) 70 wt% NaAlg/30 wt% glycerol film (control film), (b) *Film 1*, (c) *Film 2*, and (d) *Film 3*. The red non-continuous line indicates the borders of the glass slide. Photographs of Petri dishes containing *C. albicans* after 48 h incubation in the presence of (e) 70 wt% NaAlg/30 wt% glycerol film, (f) *Film 1*, (g) *Film 2*, and (h) *Film 3*. (For interpretation of the references to color in this figure legend, the reader is referred to the web version of this article.)

Fig. 8. (a) Growth curves of *C. albicans* in LB (blue line) medium containing 1 g CaAlg beads (red line), and in LB containing 1 g CaAlg/PVPI complex beads (green line). Note that in LB and in LB containing pure CaAlg beads *E. coli* growth rates were practically the same, whereas in the presence of CaAlg/PVPI complex beads no *C. albicans* growth was observed. (b) Growth curves of *E. coli* in the presence of 12 g of CaAlg beads (control experiment, red line), and CaAlg/PVPI complex beads (purple and green lines). 1 g CaAlg/PVPI complex beads was not enough to terminate the growth of the bacteria in this case (purple line). 8 g of CaAlg/PVPI complex beads was sufficient for the inhibition of bacteria growth (green line). (For interpretation of the references to color in this figure legend, the reader is referred to the web version of this article.)

in Petri dishes containing the bacterial or fungal media as seen in Fig. 7. For *E. coli* bacteria tests plastic Petri dishes were filled with Luria–Bertani agar medium. Fig. 7(a) shows the photograph of NaAlg/glycerol (control) film incubated in the heavily populated *E. coli* medium for 48 h. At the end of the 48 h period, growth of bacteria all over the Petri dish was noticed. On the other hand, as seen in Fig. 7(b–d) corresponding to Films 1, 2 and 3, there was no *E. coli* growth on the coatings, and a thin inhibition zone free from bacteria was noticed around the films (the borders of which are shown with red dashed lines) after 48 h of incubation. The antibacterial effect of the three NaAlg/PVPI films was similar and there appeared to be no significant differences in bacteria growth inhibition between the films of lowest and highest PVPI concentration. For *C. albicans* tests, plastic Petri dishes were filled with SDA medium. The antifungal activity of the same films against *C. albicans* after 48 h of incubation in Petri dishes is reported in Fig. 7(e–h), respectively. The control film (NaAlg/glycerol) (Fig. 7e) did not show any antifungal activity. On the contrary, films containing PVPI complex inhibited the growth of *C. albicans* considerably. In fact, in this case, 24 h of incubation (not shown here) was sufficient to inhibit fungal growth with similar results to those obtained after 48 h incubation. These images clearly indicate that PVPI complex is highly active against fungal growth even if it is embedded in the NaAlg matrix. The formation of an inhibition zone around all films containing PVPI complex also indicates PVPI release from the NaAlg matrix, rendering these composites not only surface active but also prohibitive for fungal growth in their vicinity.

To assess the antibacterial and antifungal activity of CaAlg/PVPI beads against *E. coli* and *C. albicans* cells in water, we performed turbidity assays by detecting the spectral absorbance values at 600 nm of the cultures dispersed in water at different times. Absorbance at 600 nm is the generally recommended wavelength for measuring solutions containing *E. coli* or other bacteria (Glover, 1985; Harley & Prescott, 2002). As shown in Fig. 8(a), the growth curves of *C. albicans* incubated with 1 g of antiseptic-free CaAlg beads are similar to the control experiment (without beads). On the other hand, the curve related to fungal growth with 1 g of NaAlg/PVPI beads clearly shows that the populations of cells do not increase at all, thus indicating cell death. Similar results were obtained with *E. coli* cultures as shown in Fig. 8(b), though a higher amount of CaAlg/PVPI beads (12 g) were needed to terminate the bacterial growth in this case. This is because the minimal bactericidal concentration (MBC) of PVPI was higher with respect to the minimal fungicidal concentration (MFC) (see Supplementary data).

5. Conclusions

Biodegradable NaAlg/PVPI composite films and Ca²⁺-cross-linked CaAlg/PVPI composite beads were prepared for the first time and their antimicrobial and antifungal activity was demonstrated. Alginate matrix enables time controlled release of PVPI, confirmed by UV spectra measurements, prolonging the well known PVPI antiseptic effect. Optical and AFM microscopy studies showed that PVPI complex forms circular micro-domains in the alginate matrix indicating micro-phase separation within the films. Detailed FTIR studies indicated that the blending process of the two materials does not cause chemical interactions thereby preserving PVPI's antiseptic properties. Moreover, cross linking NaAlg/PVPI blends with aqueous CaCl₂ solutions enabled encapsulation of PVPI complex in CaAlg beads, which allow controlled release of the antiseptic in water saturated media. The most important potential application of these materials could be their use as external and internal wound dressings with controlled drug release properties. Their advantage toward the existing materials is that they are fully biocompatible and inexpensive. During surgery such beads can be easily used in open wounds without the need to remove them and without toxicity risks. Beads can be also incorporated in field waters as cost-effective disinfectants.

Appendix A. Supplementary data

Supplementary data associated with this article can be found, in the online version, at <http://dx.doi.org/10.1016/j.carbpol.2012.09.034>.

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